

Core-Double-Shell, Carbon Nanotube@Polypyrrole@MnO₂ Sponge as Freestanding, Compressible Supercapacitor Electrode

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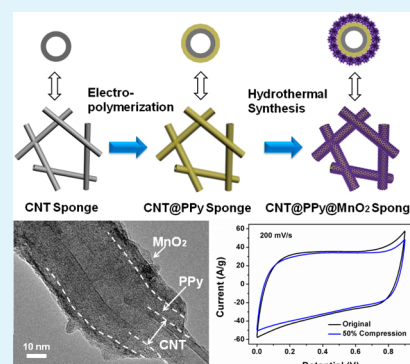
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S Supporting Information

ABSTRACT: Design and fabrication of structurally optimized electrode materials are important for many energy applications such as supercapacitors and batteries. Here, we report a three-component, hierarchical, bulk electrode with tailored microstructure and electrochemical properties. Our supercapacitor electrode consists of a three-dimensional carbon nanotube (CNT) network (also called sponge) as a flexible and conductive skeleton, an intermediate polymer layer (polypyrrole, PPy) with good interface, and a metal oxide layer outside providing more surface area. These three components form a well-defined core-double-shell configuration that is distinct from simple core-shell or hybrid structures, and the synergistic effect leads to enhanced supercapacitor performance including high specific capacitance (even under severe compression) and excellent cycling stability. The mechanism study reveals that the shell sequence is a key factor; in our system, the CNT-PPy-MnO₂ structure shows higher capacitance than the CNT-MnO₂-PPy sequence. Our porous core-double-shell sponges can serve as freestanding, compressible electrodes for various energy devices.

KEYWORDS: carbon nanotube sponge, core-double-shell, supercapacitor, flexible electrode



1. INTRODUCTION

Developing efficient energy storage systems such as supercapacitors and batteries demands rational design and controlled fabrication of bulk electrode materials.^{1–6} Combining materials with different functions to form hybrid, hierarchical structures is an effective approach toward novel-design and high performance electrodes. This strategy has been explored in supercapacitors in which binary and ternary composite electrodes have been widely studied.^{7–16} Specifically, amorphous carbon particles, carbon nanotubes (CNTs), or graphene forming conductive networks are combined with pseudo materials such as polymers and metal oxides to make a composite structure.^{17–22} Previous study has demonstrated the synergistic effect produced by integrating multiple components, which could overcome the limitation of pseudo materials (e.g., low conductivity of metal oxides) and improve specific capacitance. In this regard, carbon materials-conducting polymers,^{8,10,17,21,22} carbon materials-metal oxides,^{9,18–20} conducting polymers-metal oxides,^{11,23,24} and carbon materials-metal oxides-conducting polymers^{13,25–30} have been fabricated and investigated for supercapacitor electrodes.

Although there have been several studies on ternary electrodes involving CNTs (graphene) and two types of pseudo materials, those electrodes do not exhibit a well-defined three-dimensional (3D) configuration and the composite structure is not well controlled.^{13,25–30} Typically, CNTs were dispersed in solution and mixed with precursors for pseudo materials, and the hybrid structure was synthesized by appropriate treatments (thermal annealing).^{13,26–29} Those products are generally in powder form in which aggregation of active materials would compromise the porosity and available surface area. More importantly, because CNTs were hybridized with pseudo materials first and then assembled into electrodes, it is difficult to create a highly conductive CNT network due to the presence of a pseudo material layer between CNTs (preventing direct inter-CNT contact). The mechanical strength and flexibility are not high, and binders are usually added in order to make freestanding electrodes. In addition, pseudo materials such as polymers and metal oxides are grafted

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onto CNTs as a mixture rather than individual layers,^{25,27} making it less straightforward for mechanism study and structural optimization.

Here, we report a 3D porous hierarchical sponge electrode that can fully utilize the electrical conductivity of CNT networks and also has a synergistic effect from a combination of two pseudo materials with tailored structure. The hybrid structure is based on CNT sponges reported by our group, which have demonstrated supercapacitor properties as pristine and hybrid sponges.^{31,32} The electrode is fabricated by direct deposition of pseudo materials into the porous sponge while protecting the original interconnected CNT network with high conductivity. A polymeric and metal oxide pseudo material was selected, respectively, creating a macroscopic CNT@PPy@MnO₂ (PPy is polypyrrole) core-double-shell sponge with good flexibility (allowing severe compression without collapse). The core-double-shell skeleton is the most important feature here (also distinct from previous simple core-shell structures) that is responsible for high capacitance, and the shell sequence (PPy outside or MnO₂ outside) can be controlled to understand related mechanism and optimize electrode configuration. Also, the ternary hybrid sponges maintain stable characteristics under large degree compression with improved volume-normalized capacitance. Our core-double-shell sponges represent a possible route toward high performance electrodes with predefined configuration and controlled fabrication.

2. EXPERIMENTAL SECTION

2.1. Fabrication of CNT@PPy@MnO₂ Core-Double-Shell Sponges. CNT sponges were synthesized by chemical vapor deposition (CVD), as reported by our group previously. To fabricate the core-double-shell sponge, we first synthesized the CNT@PPy core-shell sponge by electropolymerization and then attached a MnO₂ shell through a hydrothermal process. An as-grown bulk CNT sponge block with edge size of several mm was immersed in Py/acetone solution with a concentration of 0.1 M for 0.5 h. Then, the CNT sponge with Py monomers adsorbed on the surface of each individual CNT was used as the working electrode under a potential of 0.8 V in 0.3 M NaClO₄ aqueous solution. The electropolymerization of Py was performed in a three-electrode electrochemical workstation (CHI660D instruments, Shanghai, China). A Pt wire and Ag/AgCl were used as the counter and reference electrodes, respectively. After a typical electropolymerization time of 10 minutes, the CNT@PPy core-shell sponge was obtained. The PPy loading can be controlled by the Py concentration in acetone and the absorption period. In the hydrothermal process, 0.02 mol/L KMnO₄ solution was firstly prepared and the pH value of the solution was adjusted by slowly adding sulfuric acid to pH = 1. The obtained CNT@PPy sponge was immersed in the KMnO₄ solution, further stirred for 30 s, and then transferred into a Teflon-lined stainless steel autoclave, sealed and maintained at 120 °C for 20 min. A MnO₂ shell was attached outside the PPy shell. The resulting sponge was rinsed with distilled water and then freeze-dried to maintain the porous structure.

2.2. Material Characterization. The sample morphology and structure were characterized using scanning electron microscopy (SEM, LEO 1530), transmission electron microscopy (TEM, JEOL 2010), and an X-ray diffraction (XRD) spectroscope (Bruker-D8 Discover). FTIR spectra were recorded on a FTIR system (Bruker Vector22). X-ray photoelectron spectroscopy (XPS) was performed using an ESCALAB 250Xi spectrometer. Raman spectra were recorded using a RM 2000 Microscopic Confocal Raman Spectrometer (Renishaw PLC, England) with a 633 nm laser. Thermogravimetric analysis (TGA) was conducted on a TGA Q5000 analyzer from 30 to 800 °C under air at a heating rate of 20 °C min⁻¹. Mechanical tests were carried out by a single-column static instrument (Instron 5843) equipped with two flat-surface compression stages and a 10 N load cell. The N₂ adsorption and desorption isotherms were measured at 77

K in a liquid nitrogen bath using a QUANTACHROME Autosorb-iQ analyzer. The BET specific surface area of sponges was measured. The pore volume and pore size distribution were obtained using the Barrett–Joyner–Halenda (BJH) method.

2.3. Electrochemical Measurements. The electrochemical properties of the sponge including cyclic voltammetry (CV), galvanostatic charge/discharge, and electrochemical impedance spectroscopy (EIS) were measured using a CHI660D electrochemical workstation in a three-electrode configuration. The sponge clamped by two polymeric blocks in original state (no compression) or at predefined compressive strains was used as working electrode. Platinum wire twisted around the polymeric clamp and connected to the sponge electrode was used as the current collector. A platinum wire and a saturated calomel electrode (SCE) were used as counter and reference electrodes, respectively. The CV curves were measured under different scan rates of 2–200 mV/s between 0 and 0.9 V. EIS measurements were carried out in the frequency range from 100 kHz to 0.01 Hz at open circuit potential with an ac perturbation of 5 mV. Electrochemical measurements were carried out in a 2 M KCl aqueous electrolyte at room temperature. The specific capacitance of the sponge electrode (C_s) was calculated from the CV curves and discharging curves according to eqs 1 and 2, respectively.

$$C_s = \int \frac{I}{m} dV / \nu \Delta V \quad (1)$$

$$C_s = \frac{I \Delta t}{m \Delta U} \quad (2)$$

where I is the response current (A), m is the total mass the sponge electrode (including CNTs, MnO₂, and PPy) (g), ΔV is the potential range in the CV (V), ν is the potential scan rate (mV/s), ΔU is the potential window in the discharging process, and Δt is the discharging time.

3. RESULTS AND DISCUSSION

Our method involved two hybridization steps, in which a porous CNT sponge was immersed in precursor solutions to deposit different pseudo materials sequentially in a controlled manner, as illustrated in Figure 1 (see Experimental Section for details). First, a uniform PPy layer was coated onto the CNT surface throughout the sponge by electropolymerization,

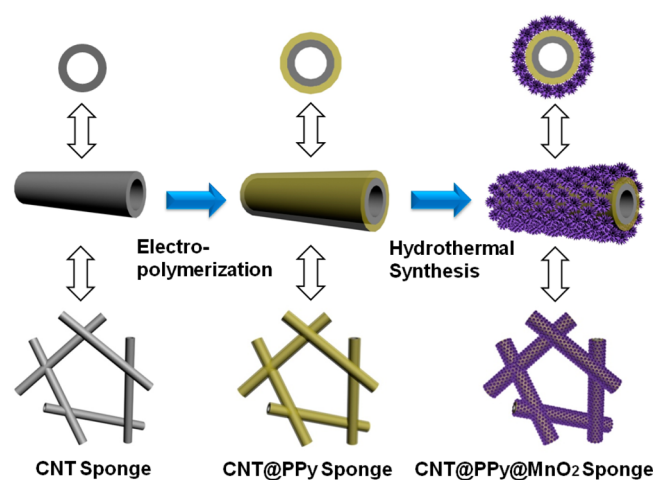


Figure 1. Schematic for the fabrication process from CNT sponge to CNT@PPy@MnO₂ core-double-shell sponge. A uniform PPy layer was firstly coated on the surface of individual CNTs by electropolymerization to produce an intermediate CNT@PPy core-shell sponge. Then, through a hydrothermal process, a MnO₂ shell was attached outside the PPy shell, leading to the formation of the core-double-shell CNT@PPy@MnO₂ sponge.

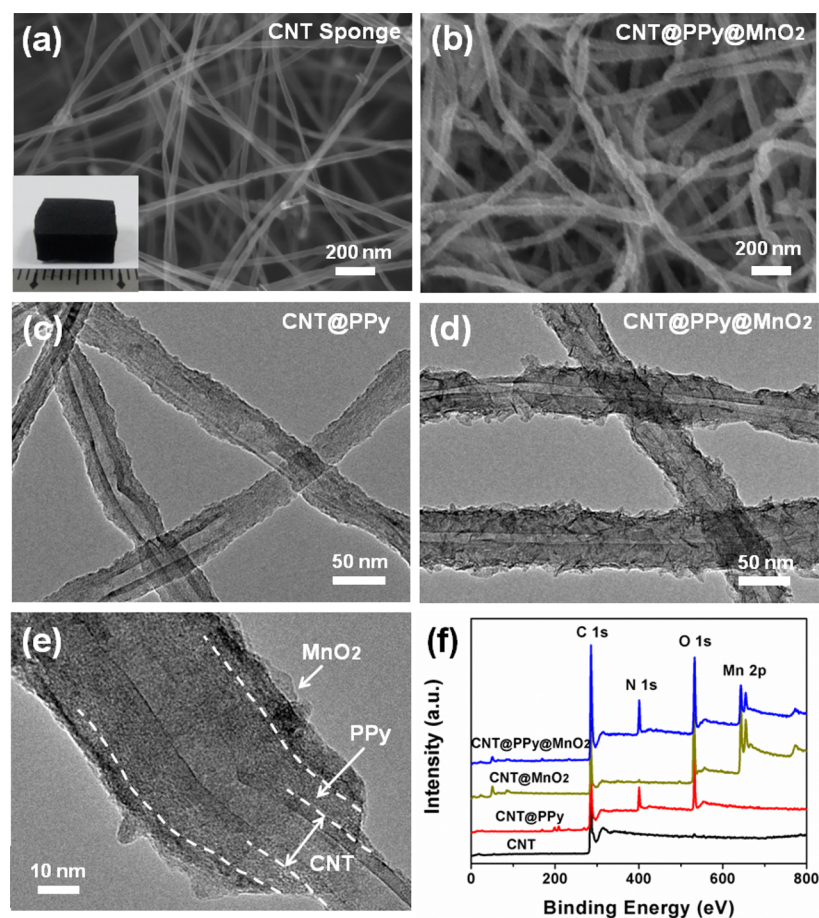


Figure 2. Characterization of CNT and core-double-shell sponges. (a) Cross-sectional SEM image of the CNT sponge. Inset shows a sponge block with edge size of several mm. (b) SEM image of the CNT@PPy@MnO₂ sponge. (c) TEM image of the CNT@PPy sponge. (d) TEM image of the CNT@PPy@MnO₂ sponge. (e) High-magnification TEM image showing the core-double-shell structure. (f) Surface scanning XPS full spectrum showing signals from Mn, O, N, and C elements in the CNT@PPy@MnO₂ sponge, compared with other sponges.

forming a binary CNT@PPy sponge. Second, a MnO₂ layer was attached to the PPy shell by a low temperature hydrothermal process, resulting in a ternary CNT@PPy@MnO₂ sponge with increased hierarchy. Because all CNTs were completely immersed in precursor solutions, the PPy and MnO₂ layers wrapped the entire surface of each nanotube nicely so that a core-double-shell structure formed. Our core-double-shell sponges have three distinct features compared to previous hybrid materials: (1) Uniform PPy and MnO₂ layers with controlled thickness and mass loading can be coated throughout the sponge. A reversed shell sequence, CNT@MnO₂@PPy also can be obtained by first depositing MnO₂ (as inside shell) and then PPy (outside). (2) The original CNT network (including the inter-CNT junctions) was embedded within the pseudo material layers so that the electrical conductivity is preserved at maximum. (3) At a relatively thin coating thickness, the resulting core-double-shell sponge could maintain much porosity and flexibility of original CNT sponges. The CNT@MnO₂@PPy sponge can be compressed to large strains due to the presence of a robust CNT skeleton. These are favorable structural factors for supercapacitor applications.

We have characterized the structure of CNT and CNT@PPy@MnO₂ sponges by SEM and TEM. An as-synthesized CNT sponge (inset of Figure 2a) consists of a 3D network of randomly overlapped nanotubes, creating an open porous structure (Figure 2a). These are multi-walled CNTs with

diameters of 20–50 nm and lengths up to tens of micrometers.³³ Such porous structure is favorable for depositing active materials by electropolymerization and hydrothermal processes, as the precursor solution could easily infiltrate into the inner portion of a bulk sponge through the empty space between CNTs. After grafting PPy and MnO₂, the CNTs are wrapped by a uniform shell and the average outer diameter has increased from about 25 nm (clean nanotubes) to 55 nm (with PPy and MnO₂) (Figure 2b). The original CNT network is well preserved, and the CNT@PPy@MnO₂ sponge remains highly porous. SEM images of a CNT@PPy and CNT@MnO₂ sponge show distinct morphologies after coating PPy or MnO₂, respectively (Supporting Information, Figure S1).

TEM images of products at different stages (CNT@PPy and CNT@PPy@MnO₂, respectively) clearly reveal the structure evolution during hybridization. For the CNT@PPy core-shell sponge, there is a PPy layer (thickness of 10–20 nm) coated on the surface of individual CNTs (Figure 2c). In comparison, CNT surfaces within the CNT@PPy@MnO₂ sponge become rough due to the outside MnO₂ layer (Figure 2d), which usually exists as protruding nanosheets as reported before.^{25,28} The diameters of the CNT@PPy@MnO₂ units are further increased to 60–80 nm. An enlarged view of a particular region clearly shows the core-double-shell structure consisting of a CNT inside, a PPy layer in the middle, and a MnO₂ layer at the outside; both shells have thicknesses of about 10 nm (Figure

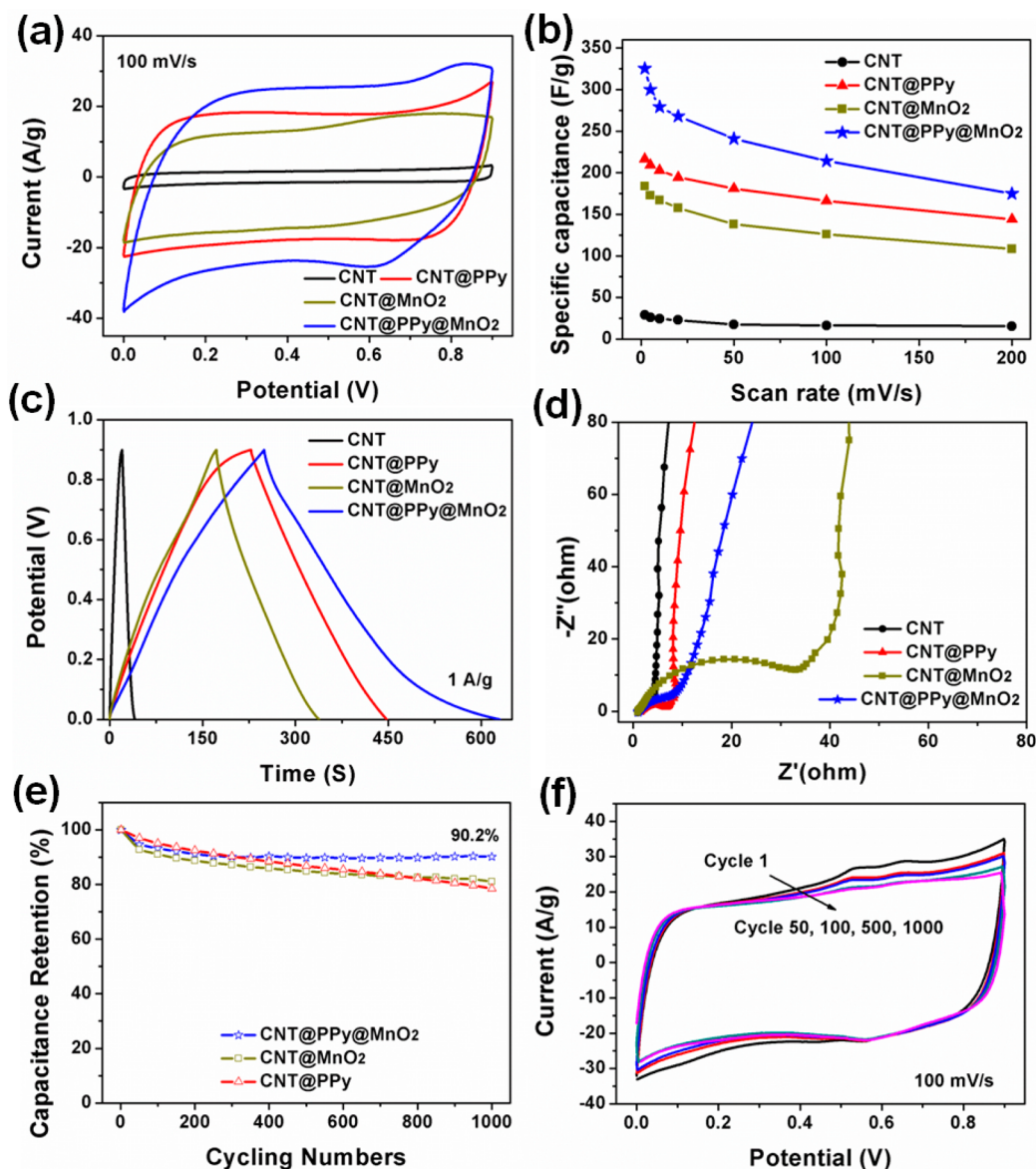


Figure 3. Supercapacitor application of CNT@PPy@MnO₂ core-double-shell sponges. (a) CV curves of the CNT, CNT@PPy, CNT@MnO₂, and CNT@PPy@MnO₂ sponges at a scan rate of 100 mV/s. (b) Calculated specific capacitances of the four types sponges at different scan rates. (c) Galvanostatic charge/discharge curves of the four sponges at a current of 0.5 A/g. (d) Nyquist plots of the EIS for the four sponges. (e) Cycling tests showing a better stability of CNT@PPy@MnO₂ sponge than CNT@PPy and CNT@MnO₂ sponges after 1000 charging and discharging cycles at 100 mV/s. (f) CV curves of cycle 1st, 50th, 100th, 500th, and 1000th for the CNT@PPy@MnO₂ sponge.

2e). The CNT and its two shells have good interfaces (without cracks or holes), which ensures efficient charge transport along the radial direction from the outer shells to the CNT. Due to the strong adhesion of PPy to CNT by electropolymerization and the chemical connection between PPy and MnO₂, the PPy layer also serves as an intermediate layer for anchoring MnO₂ nanosheets, which otherwise could easily detach from the smooth CNT surface. Strong adhesion of pseudo materials to the CNT network is important for making stable supercapacitor electrodes. In addition, the rough MnO₂ shell also leads to increased specific surface area (108 m²/g) in the hybrid sponge compared to the original CNT sponge (80 m²/g), as well as increased percentage of pores with sizes in the range of 1.5–100 nm (accessible by liquid electrolyte) (Supporting Information, Figure S2).

Surface elemental analysis was carried out by XPS on the original and different hybrid sponges. The full XPS spectrum of

a CNT@PPy@MnO₂ sponge reveals signals from Mn, O, N, and C elements, while there is only a C1s peak of an original CNT sponge (Figure 2f). Consistently, we observe N1s peak in a CNT@PPy sponge and Mn 2p peak in CNT@MnO₂. The high-resolution Mn 2p core level spectra of CNT@PPy@MnO₂ and CNT@MnO₂ sponges show that Mn 2p_{3/2} and Mn 2p_{1/2} (having binding energies centered at 642.5 and 654.3 eV, respectively, with a spin energy separation of 11.8 eV) are in agreement with the reported data for MnO₂ (Supporting Information, Figure S3a).^{29,30} We also studied the hybrid sponge structure by other characterization techniques including FTIR, XRD, Raman spectra, and TGA. The FTIR test results further verify the presence of PPy in the CNT@PPy@MnO₂ sponge. The peaks at ca. 1500 and 1455 cm⁻¹ in the CNT@PPy@MnO₂ sample can be attributed to C–C and C–N stretching vibrations in the pyrrole ring, respectively (Supporting Information, Figure S3b). The CNT@PPy@MnO₂ sponge

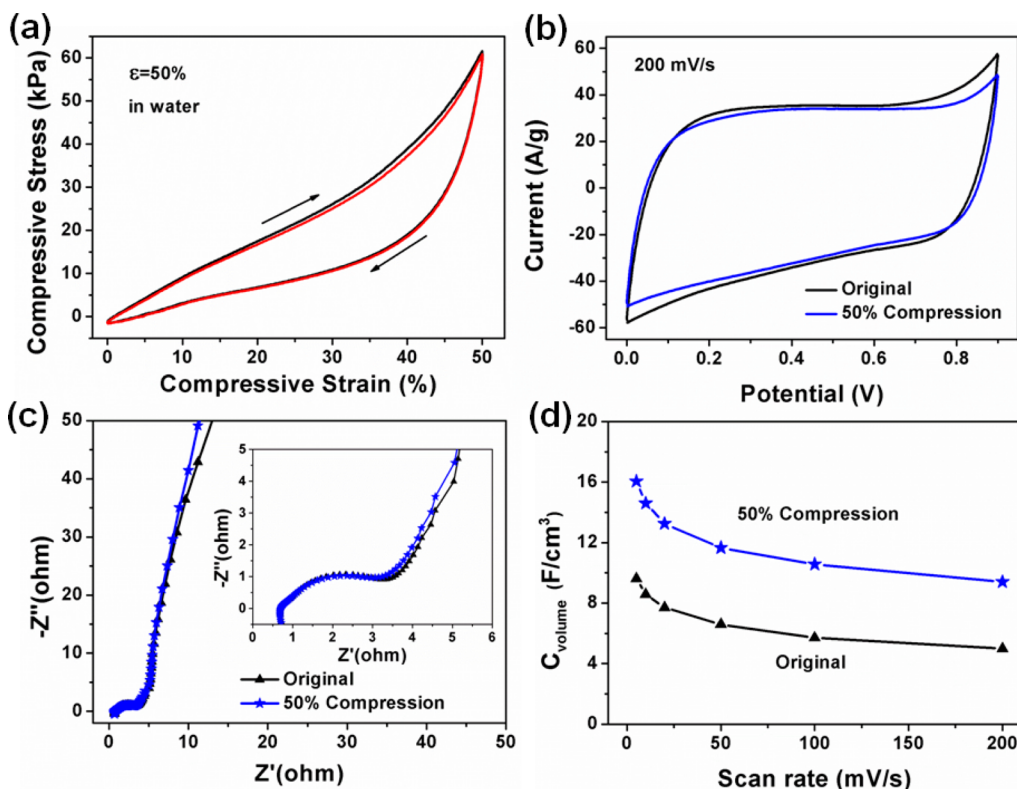


Figure 4. Supercapacitor performance under compression. (a) Mechanical testing (compressive stress-strain curves) of a CNT@PPy@MnO₂ sponge immersed in water and compressed to a strain of 50% for two cycles. The unloading curve shows elastic recovery. (b) CV curves of a CNT@PPy@MnO₂ sponge in original and compressed states ($\epsilon = 50\%$) at 200 mV/s. (c) Nyquist plots of the EIS for the sponge in original and compressed states. Inset is the magnification of Warburg semicircles from the spectra. (d) Volume-normalized capacitances (C_{volume}) of the sponge in original and compressed states at different scan rates.

shows a characteristic peak at 37.5° (although relatively weak) in the XRD pattern, indicating the presence of MnO₂ probably in an amorphous nature (Supporting Information, Figure S3c), which is favorable for supercapacitor applications.³⁴ Raman and TGA results also confirm the presence of MnO₂ (Raman peak at $480\text{--}680\text{ cm}^{-1}$) and PPy (causing early weight loss when heated to about 100°C) (Supporting Information, Figure S3d,e).

In order to evaluate the electrochemical performance of the as-prepared freestanding CNT@PPy@MnO₂ core-double-shell sponge, cyclic voltammetry (CV), galvanostatic charge/discharge, and electrochemical impedance spectroscopy (EIS) measurements in a three-electrode system were employed. The electrolyte is 2M KCl aqueous solution. Figure 3a compares the CV curves of the CNT@PPy@MnO₂, CNT@PPy, CNT@MnO₂, and CNT sponge electrodes at a same scan rate of 100 mV/s. The mass content of MnO₂ and PPy in the hybrid sponges are $\sim 40\%$ and $\sim 20\%$ wt %, respectively. Detailed mass content for the hybrid sponge samples is shown in the Supporting Information, Table S1. The CV curves exhibit good rectangular and symmetric shapes, suggesting fast reversible Faradic reactions and ideal capacitive behavior. The PPy and MnO₂ functionalization leads to higher current density and enlarged area enclosed by the charging/discharging curves, with a specific capacitance (at 100 mV/s) increasing from 16.4 F/g (for CNT sponge) to 126.1 F/g for CNT@MnO₂, 166.3 F/g for CNT@PPy, and 214.2 F/g for CNT@PPy@MnO₂ sponge, respectively. The specific capacitance is calculated on the basis of the total hybrid sponge weight (CNTs and the active materials). Furthermore, the CNT@PPy@MnO₂ sponge shows

a stable shape as the scan rate increased from 2 to 200 mV/s (Supporting Information, Figure S4). It reveals good rate performance, especially at large charge/discharge current densities. We have optimized the MnO₂ mass loading by controlling the hydrothermal synthesis period. The optimized synthesis period is 20 min with corresponding MnO₂ loading of ca. 39.1 wt % (Supporting Information, Figure S5). The highest capacitance is ca. 325 F/g obtained in the CNT@PPy@MnO₂ sponge at a scan rate of 2 mV/s, in contrast to 29.3 F/g for CNT sponge, 183.7 F/g for CNT@MnO₂, and 216.3 F/g for CNT@PPy, respectively (Figure 3b). This performance improvement is attributed to the synergistic effect of the high pseudo-capacitance of the conformal PPy and MnO₂ coating and the porous structure of CNT network. In the CNT@PPy@MnO₂ sponge, the inter CNTs provide a porous structure and a conductive network. The MnO₂ layer is mainly responsible for pseudo reactions; the middle PPy layer can improve the interaction between the outer MnO₂ layer and the inter CNTs, reduce the interface resistance, and improve the charge transfer. Figure 3c shows the galvanostatic charge/discharge curves at a constant current density of 1 A/g. The symmetrical triangle curves indicate good reversibility of the CNT@PPy@MnO₂ core-double-shell sponge electrode. Additionally, negligible voltage drop (IR drop) can be seen from the curve, indicating a really low internal resistance which is important for energy electrodes. We have also carried out the two-electrode configuration test by assembling the core-double-shell sponges into a prototype two-electrode symmetrical supercapacitor in 2M KCl aqueous electrolyte (Supporting Information, Figure S6). The sponge shows good rate performance, even under a

high scan rate of 1000 mV/s and a high charge/discharge current density of 50A/g. The highest specific capacitance of the electrode is 305.9 F/g at a scan rate of 2 mV/s (Supporting Information, Figure S6c). The maximum energy density and power density are 8.6 Wh/kg and 16.5 KW/kg, respectively (Supporting Information, Figure S6d).

The EIS test is conducted for further characterizing the CNT@PPy@MnO₂ electrode. Figure 3d shows the resulting Nyquist plots of the EIS spectra for the CNT@PPy@MnO₂, CNT@PPy, CNT@MnO₂, and CNT sponge electrodes. All spectra are composed of a typical semicircle in the high frequency region and a straight line at low frequency. The intercept of the semicircle on the real axis at high frequency represents the equivalent series resistance (R_s), and the diameter of the semicircle corresponds to the charge-transfer resistance (R_{ct}) of the electrode.²⁰ The R_s of the hybrid sponge electrodes are approximate ($\sim 1.0 \Omega$). However, the CNT@PPy@MnO₂ sponge electrode shows an obvious decrease of R_{ct} than that of the CNT@MnO₂ sponge electrode. This indicates that the middle PPy shell can improve the interaction between the outer MnO₂ nanosheets shell and the CNTs in the CNT@PPy@MnO₂ sponge, rendering electron transport from MnO₂ to CNTs more efficient. Cycling stability of the CNT@PPy@MnO₂ electrode has been investigated at a scan rate of 100 mV/s for 1000 charging/discharging cycles. The CNT@PPy@MnO₂ electrode shows a capacitance retention of 90.2%, which is higher than CNT@MnO₂ (81.1%) and CNT@PPy (78.5%) (Figure 3e). The CV curves become almost identical after about 500 charging and discharging cycles, indicating good stability of our CNT@PPy@MnO₂ sponge electrode (Figure 3f).

To further demonstrate the structure advantage of CNT@PPy@MnO₂, a CNT@MnO₂@PPy sponge with reversed shell sequence was fabricated by changing the order of the two synthesis processes. The CNT@MnO₂@PPy sponge also has a core-double-shell configuration but with PPy outside. We compared the supercapacitor performance of the two ternary hybrid sponges with similar PPy and MnO₂ loadings (Supporting Information, Figure S7). The CNT@MnO₂@PPy sponge shows a lower capacitance (241 F/g) than the CNT@PPy@MnO₂ sponge (325 F/g) (Supporting Information, Figure S7c). In particular, the R_{ct} of the CNT@MnO₂@PPy sponge is much higher than the CNT@PPy@MnO₂ sponge (Supporting Information, Figure S7d). We also compare another set of ternary hybrid sponge samples with more MnO₂ loading (ca. 58 wt %). The CNT-PPy-MnO₂ structure sequence also shows better performance than the CNT-MnO₂-PPy sequence (Supporting Information, Figure S8). This demonstrates that, in our CNT-PPy-MnO₂ ternary hybrid sponge, the core-double-shell structure design will affect the electrochemical performance.

Besides the improvement of specific capacitance by forming PPy and MnO₂ double shells, the CNT@PPy@MnO₂ sponge also can work under large degree compression. Figure 4a shows that the sponge can be immersed in water and compressed elastically. Two loading and unloading cycles show full thickness (volume) recovery by water absorption, which accompanies the fast sponge expansion to the original volume owing to the robust and elastic CNT network. We measured the supercapacitor performance of the same sponge in the original state and under *in situ* compression performed by specially designed clamps (Supporting Information, Figure S9). The sponge exhibits similar CV characteristics at a compressive

strain of 50% compared with the original state (Figure 4b). Even when its volume has been greatly reduced, the sponge still retains more than 90% of the original specific capacitance. Since the 3D CNT network sustains an open porous and stable structure even under high compressive strain, liquid electrolyte can readily infiltrate and access the inner part of the compressed sponge, which is important for ion diffusion and minimizing capacitance loss. The galvanostatic charge/discharge tests of the original and compressed sponge are consistent with their CV results (Supporting Information, Figure S10). Under a high compressive strain (50%), the charge/discharge curve of the sponge maintains a symmetrical shape. The EIS of the CNT@PPy@MnO₂ sponge is shown in Figure 4c. The CNT@PPy@MnO₂ sponge shows similar Nyquist plots in original and compressed states ($\epsilon = 50\%$), indicating a stable ESR ($\sim 0.70 \Omega$) under compression. We further calculate the volume-normalized specific capacitance (C_{volume}), and in this case, the compressed sponge shows significantly improved C_{volume} because of large volume reduction. For example, at a scan rate of 5 mV/s, the C_{volume} value reaches 16.1 F/cm³ under 50% compression, showing ca. 67% increase compared to the original sponge (9.6 F/cm³) (Figure 4d). This result indicates it is possible to fabricate a compressible supercapacitor with enhanced volume-normalized capacitance.

4. CONCLUSION

In summary, we demonstrated a three-component CNT@PPy@MnO₂ sponge electrode with a distinct core-double-shell structure. Pseudo material layers with controlled thickness and shell sequence were coated uniformly throughout the CNT sponge, maintaining the original CNT network and interconnection. Synergistic effect from respective components (PPy and MnO₂) plus the porous conductive CNT network offers the hybrid sponge a superior electrochemical performance. The 3D CNT network can significantly reduce the internal resistance of the electrode and maintain the structure stability. Porous CNT@PPy@MnO₂ sponges can serve as freestanding, compressible electrodes for supercapacitors and other energy devices. Our core-double-shell structure may be applied to other electrochemical materials for design and fabrication of hybrid and hierarchical electrodes.

■ ASSOCIATED CONTENT

Supporting Information

The characterizations of the core-double-shell sponge, including XRD, Raman, TGA, nitrogen sorption isotherms, and pore size distribution. Supercapacitor performance comparison between the CNT@MnO₂@PPy and CNT@PPy@MnO₂ sponges. The supercapacitor performance of the CNT@PPy@MnO₂ sponge by the two-electrode configuration test. This information is available free of charge via the Internet at <http://pubs.acs.org/>.

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Notes

The authors declare no competing financial interest.

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